

# Production of Green energy via Water Splitting mechanism by Mn-doped cobalt ferrites [Co<sub>1-x</sub>Mn<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub>] based hydroelectric cells

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Volume 1, Issue 2, March 2024

Received: 11 January, 2024; Accepted: 14 February, 2024

DOI: <https://doi.org/10.63015/2h-2419.1.2>

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**Abstract:** In our current study, we have investigated the impact of manganese (Mn) doping on the structural, magnetic properties, and hydroelectric cell properties of cobalt ferrite with chemical formula [Co<sub>1-x</sub>Mn<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub>, x=0.00, 0.125]. The solid-state approach has been utilized to synthesize the material. XRD measurements revealed the formation of an inverse spinel cubic structure with the space group Fd-3m. FTIR spectra hold a shift in the octahedral and tetrahedral bands with the increase in Mn content in cobalt ferrites. The subsequent rise in ferroelectric behavior has been observed with the net maximum saturation polarization of 1.08 μC/cm<sup>2</sup>. The variation in the permittivity values has been also studied with the increase in Mn content for dry as well as wet hydroelectric cells. The impedance measurements helped us understand the contribution of grains/grain boundaries in the prepared ferrites. The enhanced value of magnetization has been noticed from 48.258 emu/g to 73.089 emu/g upon doping of Mn ions in cobalt ferrite spinel structure lattice. The Mn doped CFO exhibited enhanced performance in hydroelectric cells which demonstrates the great ionic diffusion mechanism with the incorporation of Mn ions. The obtained output makes Mn doped CFO a suitable material for studying microelectronic and hydroelectric cell applications. The purpose of this study is to enhance both ferroelectric as well as ferromagnetic behavior along with the improved value of current in hydroelectric cells.

**Keywords:** Ferroelectrics, Multiferroics, Hydroelectric cells, Ferromagnetism, Dielectrics.

**1. Introduction:** Energy storage has now become a big concern in the society because the consumption of energy is increasing day by day. The evolution of toxic gases by the energy released from fossil fuels is degrading the environment. So, there is a high urge to develop those devices that can generate clean and green energy. The generation of green energy by the hydroelectric cells using a few water droplets has been proven to be a cheap, clean, and eco-friendly technique [1]. Water dissociation has been obtained at room temperature by oxygen-deficient materials without the usage of any electrolytes/ acids/ alkalis, etc [2]. Hydroelectric cells have been already tailored using various metal oxides like TiO<sub>2</sub>, SnO<sub>2</sub>, MgO, ZnO-CuO, Fe<sub>2</sub>O<sub>3</sub>, and Al<sub>2</sub>O<sub>3</sub> [3-8]. It has been rigorously studied that doping of

external elements in ferrite or oxides generates oxygen vacancies in the lattice. Doping of Mg in Al<sub>2</sub>O<sub>3</sub> has decreased the internal resistance of oxide and produced the maximum power density of 3.37 mW/cm<sup>2</sup> [9]. A big amount of industrial waste called red mud has been fabricated into a clean and green source of energy (hydroelectric cells) [10].

CoFe<sub>2</sub>O<sub>4</sub> is an interesting magnetoelectric multiferroic material having a transition temperature (T<sub>C</sub>) of ~580K. Its transition temperature can be tuned below or above the room temperature by changing the Fe content in Fe<sub>2-x</sub>Mn<sub>x</sub>O<sub>4</sub> (at x =0.125 T<sub>C</sub> was 900K) and processing conditions [11]. In contemporary understanding, magnetic materials are classified into various categories, including paramagnetic, diamagnetic, ferromagnetic,

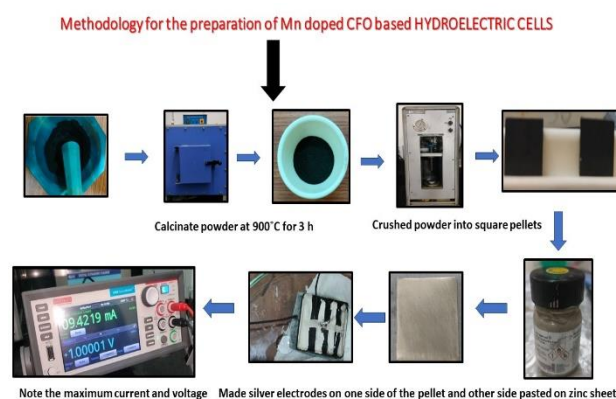
ferrimagnetic, antiferromagnetic, superparamagnetic, and spin glass. Cobalt ferrite nanoparticles have an inverse spinel structure.  $\text{Co}^{2+}$  and  $\text{Fe}^{3+}$  occupy either tetrahedral or octahedral interstitial spaces in this  $\text{O}^{2-}$  form FCC tight packing. Half of the  $\text{Fe}^{3+}$  ion and  $\text{Co}^{2+}$  ions are found in the octahedral sites of this inverse spinel cobalt ferrite structure, while the remaining  $\text{Fe}^{3+}$  ions are found in the tetrahedral sites. Cobalt ferrite features closed-packed spinel arrangements with 32 oxygen ions from unit cells and a secondary phase cubic spinel crystal structure with space group symmetry  $\text{Fd-3m}$ . The unit cell of the CFO contains a total of 56 atoms. There are 64 tetrahedral sites and 32 octahedral sites in the unit cell containing 32 oxygen ions [12].

It has been observed that Mn doped CFO has decreased the resistance of pure CFO, thereby initiating the fast movement of ions contributing towards improved current [13]. Many findings have been already reported on the storage efficiency of Mn doped CFO. Singh et al. reported the manganese doped ferrite and PANI composite for energy storage electrode material for supercapacitor applications [14]. Milan et al. found increased transport and magnetic properties of CFO upon doping with manganese ion [15]. Fiaz et al. showed the anticancer, antibacterial, and antidiabetic applications of Mn doped CFO [16].

Amongst all available ferrites, the research on manganese (Mn) doped cobalt ferrites (CFO) based hydroelectric cells (HEC) has not been reported yet. Here, we will report the maximum current and voltage produced in Mn doped CFO based HEC.

**2. Methodology:** The solid-state synthesis route has been followed to prepare the highly porous Mn doped CFO with the chemical composition  $[\text{CoFe}_2\text{O}_4, \text{CoMn}_{0.125}\text{Fe}_{1.875}\text{O}_4]$  has been shown in Fig 1. The stoichiometric amount of  $(\text{Co}_3\text{O}_4, \text{Loba Chemie})$  and  $(\text{Fe}_2\text{O}_3, \text{Fisher Scientific})$  were taken and grounded together in an agar pestle mortar in the acetic medium for 3-4 h constantly. Then, the obtained powder was calcined at  $900^\circ\text{C}$  for

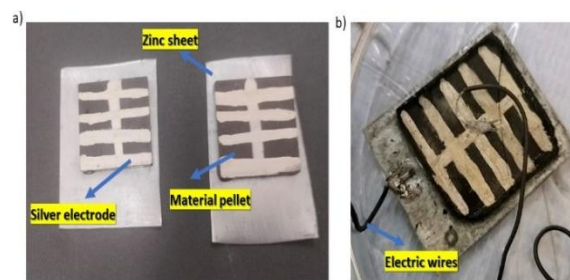
3 h. Obtained calcined powder was further sintered at  $50^\circ\text{C}$  above the calcined temperature for 4 h. Then, the powder was again grounded for 1 h to obtain fine particles.



hydroelectric cells.

### Preparation of Hydroelectric cells (HEC):

The obtained sample powder was mixed with PVA binder and then transformed into square pellets of dimensions  $2\text{ cm} \times 2\text{ cm}$  and thickness of 2mm using a hydraulic presser machine as shown in Fig 2. The square pellets were then hardened by sintering at  $250^\circ\text{C}$  for 2 h to evaporate the PVA binder and to obtain homogeneity in the samples. Silver paint lines were drawn in a comb pattern on one face of all the pellets. A zinc sheet was pasted on the other side of each of the pellets. Finally, electrical contacts were applied to all the pellets to develop them in the form of fully functional HECs.



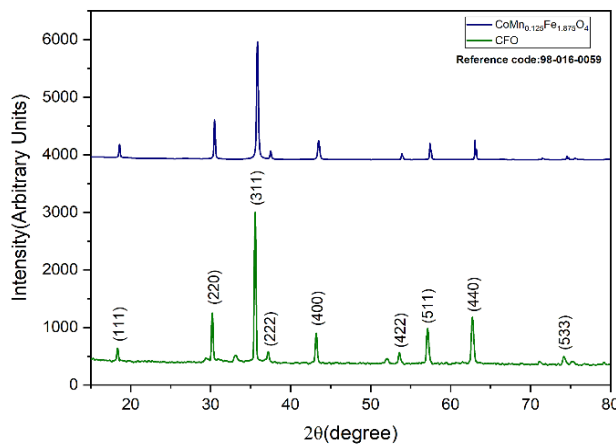
**Fig 2. Representation of (a) CFO and (b) Mn-CFO based hydroelectric cells.**

**3. Characterization Techniques:** The XRD measurements of Mn doped CFO were carried out using Bruker D8 Advance X-Ray diffractometer with  $\text{Cu-K}\alpha$  radiations ( $\lambda = 1.5406\text{\AA}$ ) in  $2\theta$  range extending from  $10^\circ$  to

90° with a scanning speed of 1°/min. The FTIR spectra were obtained using Perkin Elmer for measurements in the range of 4100-400 cm<sup>-1</sup> for solid samples. The surface morphology of the samples was done using the FESEM-EDX Zeiss instrument. Dielectric, Conductivity, and Impedance measurements of all samples were carried out by using a Nova Control Technology Impedance analyzer. The magnetic studies (M-H loop) were recorded using an ADE-EV9 VSM (vibrating sample magnetometer) instrument under a magnetic field of 1 Tesla obtained at room temperature. The V-I measurements of the HEC have been performed using a Keithley 2450 source meter.

#### 4. Results and Discussion

**4.1. Morphological analysis:** Fig 3. represents the XRD pattern of pure cobalt ferrite (CFO) and manganese doped cobalt



**Fig 3. X-ray diffraction data of CFO and CoMn<sub>0.125</sub>Fe<sub>1.875</sub>O<sub>4</sub> recorded within the 2θ region of 10°-80°; fitted with Fd-3m space group cubic crystal structure.**

ferrites (Mn CFO) recorded using a Panalytical X' Pert Pro MRD diffractometer using Cu K<sub>α</sub> radiation (λ= 1.5406Å) obtained at room temperature. The major peaks have been listed as (111), (220), (311), (220), (400), (422), (511), (440), and (533) which are

characteristics of a cubic spinel structure with Fd-3m space group. The obtained diffraction peaks are well matched with the standard JCPDS card number: **98-016-0059** for CoMn<sub>x</sub>Fe<sub>2-x</sub>O<sub>4</sub>. [17].

Table 1 lists the structural parameters that were determined from the XRD pattern, including crystallite size, lattice parameter, interplanar spacing, X-ray density, bulk density, and porosity (%). The formulae for the calculation of these parameters are

$$\text{Lattice parameter} = d\sqrt{h^2 + k^2 + l^2} \quad (1)$$

$$\text{X-ray density (D}_x\text{)} = \frac{8M}{Na^3} \quad (2)$$

$$\text{Bulk density (D}_b\text{)} = \text{Mass of pellet/ Volume of pellet} = m/\Pi r^2 L \quad (3)$$

$$\text{Porosity (\%)} = 1 - D_b/D_x \quad (4)$$

Using the W-H plot equation, the average crystallite size (D) and induced microstrain have been calculated as given by equation (5)

$$\beta \text{Cos}\theta = (4 \text{Sin}\theta)\epsilon + \frac{K\lambda}{D}, \quad (5)$$

where β represents full width at half maximum (FWHM), θ is the angle of strongest intensity peak, ε is the micro-strain, and λ is the wavelength of Cu-Kα (1.54Å) radiation used respectively. The average crystallite size for the prepared MnCFO nanoparticles has been found around 55.69 nm.

It has been observed from Table 1, that the porosity % has increased from 56.75 % to 61.8% with the incorporation of Mn ions in CFO. The doping of foreign dopant atom (Mn) in the pure CFO leads to the formation of defects resulting in a large value of microstrain [18]. The substitution of dopant ions also creates dislocation in the lattice thereby contributing towards the defects. These defects consist of oxygen vacancies which are

**Table 1. Useful Parameters calculated from the XRD plots of Manganese doped cobalt ferrites (MnCFO)**

Sample	Crystallite size (nm)	Strain (× 10 <sup>-3</sup> )	Lattice parameter (Å)	Volume (Å <sup>3</sup> )	Interplanar spacing (Å)	Bulk density (g/cm <sup>3</sup> )	X-ray density (g/cm <sup>3</sup> )	Porosity (%)
CoFe <sub>2</sub> O <sub>4</sub>	51.16	2.13	8.383	589.112	2.527	2.99	4.506	33.644
CoFe <sub>2-x</sub> Mn <sub>x</sub> O <sub>4</sub>	55.69	0.76	8.385	589.534	2.528	2.91	4.458	34.724

mainly responsible for the interaction of water molecules with the lattice.

#### 4.2. FTIR spectroscopy:

Fourier transform infrared (FTIR) spectroscopy was used to examine the structure and cation distribution between the tetrahedral and octahedral lattice sites in the inverse spinel ferrite, All spinel structures

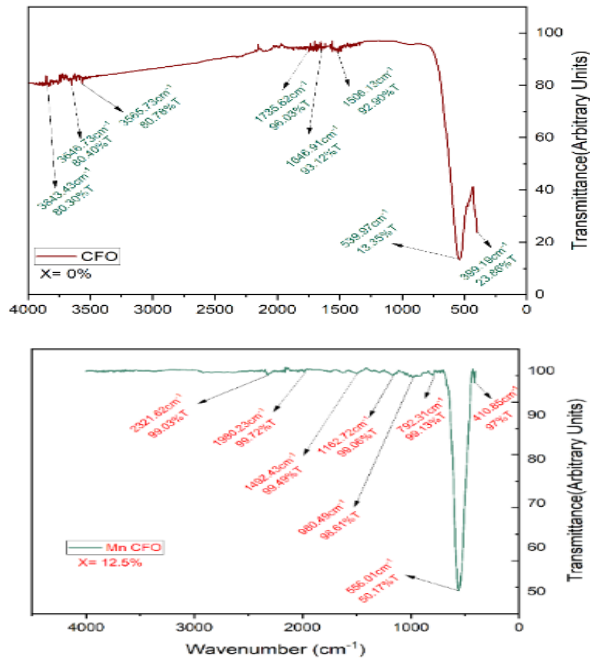


Fig 4. FTIR spectra of (a) CFO (b) Mn-CFO based ferrites.

contain two primary metal-oxygen bands that can be seen in the IR spectrum. Fig 4. shows the FTIR spectrum of pure as well as Mn-doped cobalt ferrite. The band  $\nu_1$  for pure and Mn-doped at ( $539.97\text{ cm}^{-1}$  &  $556.01\text{ cm}^{-1}$ ) arises in the range of  $500\text{-}600\text{ cm}^{-1}$  due to tetrahedral complexes and  $\nu_2$  at  $399\text{ cm}^{-1}$  &  $410.85\text{ cm}^{-1}$  is due to octahedral complexes [19]. In cobalt ferrite and cobalt ferrite doped with Mn, the O-H bands may be shown to be

stretched at a distance of  $1506.13\text{ cm}^{-1}$  and bent at a distance of  $3843.43\text{ cm}^{-1}$ , respectively. Small particle sizes can cause line broadening and vibrational mode overlap, making it challenging to observe certain bands. The characteristic band  $\nu_1$  of spinel ferrites changes to a higher frequency region as the Mn-substitution grows. This is because manganese has a lighter atomic mass than cobalt. Consequently, the frequency of metal-oxygen stretching will rise [20].

#### 4.3. Morphological Studies:

SEM-EDX micrographs of the sintered pellets

Table 2. Atomic and weight % obtained for manganese doped cobalt ferrites (Mn-CFO).

of  $\text{CoFe}_2\text{O}_4$ ,  $\text{CoMn}_{0.125}\text{Fe}_{1.875}\text{O}_4$  have been recorded at room temperature and are shown in Fig 5. The surface morphology is observed

Elements	Weight %	Atomic %
O K	21.97	50.07
Mn K	0.65	0.43
Fe K	47.42	30.96
Co K	29.96	18.53

to be quite uniform in both samples. Energy Dispersive X-ray (EDX) micro-analysis is a technique used to determine the elemental composition of a sample. The average grain size of the Mn doped spinel ferrite nanoparticles has been calculated using ImageJ software and found to be near around 69 nm. It provides qualitative information about the presence of different elements like Mn, Co, O, and Fe in the samples and allows for quantitative analysis by determining the

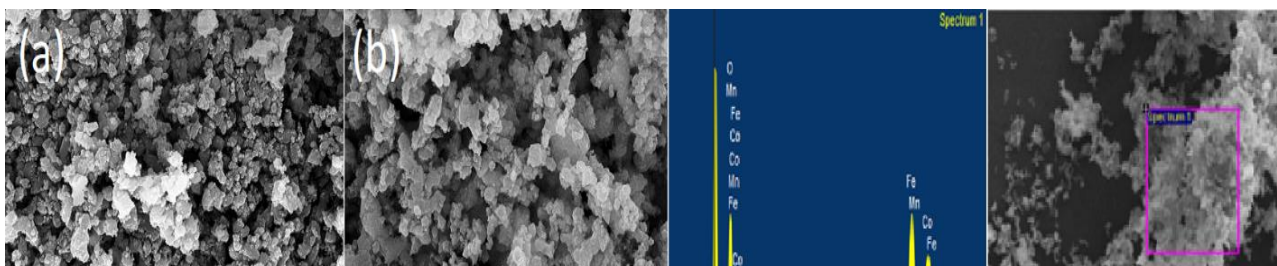
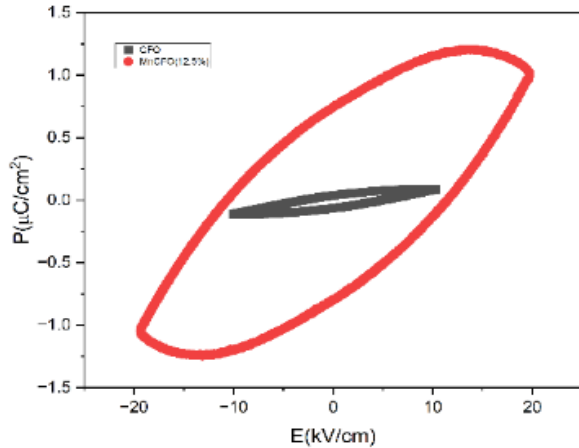


Fig 5. SEM micrograph and compositional study of (a)  $\text{CoFe}_2\text{O}_4$  (b)  $\text{CoMn}_{0.125}\text{Fe}_{1.875}\text{O}_4$ , sample at 50 k magnification and 20 kV power.



relative abundance or concentration of elements in the sample [21].

**4.4. Comparison of Ferroelectricity:** Fig 6. exhibits the variation of polarization against the electric field ( $E=10$  KV/cm) for pure as well as Mn doped cobalt ferrites obtained at

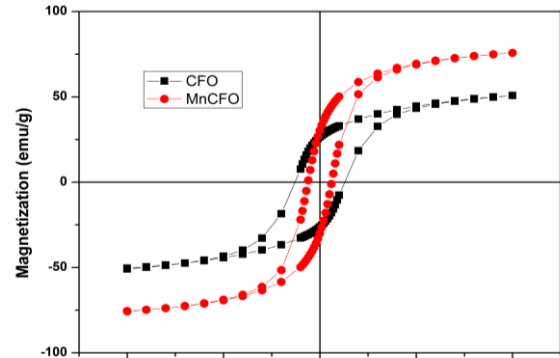


**Fig 6. Variation of Polarization (a)  $\text{CoFe}_2\text{O}_4$  (b)  $\text{CoMn}_{0.125}\text{Fe}_{1.875}\text{O}_4$ , sample with electric field obtained at RT.**

room temperature and the applied frequency at 50 Hz. The material appears to be lossy and polycrystalline based on hysteresis loops [22]. It has been observed that the net saturation polarization has been increased to  $1.08\mu\text{C}/\text{cm}^2$  with very small doping of Mn in CFO. The value of net remnant polarization has been also increased upon the addition of Mn in CFO.

**4.5. Ferromagnetic Studies:** Field-dependent magnetic properties of Mn doped CFO has been shown in Fig 7. as obtained at room temperature. It has been observed that the addition of Mn in CFO caused a vigorous rise in the values of saturation magnetization ( $M_s$ ) from 48.258 emu/g to 73.089 emu/g. Table 3. exhibits the decrease in coercivity and the subsequent rise in the magnetic properties that occurred for Mn doped CFO. The exclusive rise in the value of  $M_s$  is mainly due to the difference in the magnetic moments of the ions present at the A-site and the ions at the B-site. The doping of  $\text{Mn}^{2+}$  ion caused the replacement of B-site ions and thus altered the magnetic moment of ions present at B-site. Hence, due to the change in the magnetic moment, the net value of ( $M_s$ ) got intensively

increased [23]. The hysteresis loop has occurred at RT showing that material is ferrimagnetic. The magnetic characteristics of



**Fig 7. M-H loop of cobalt ferrite & manganese doped (a)  $\text{CoFe}_2\text{O}_4$  (b)  $\text{CoMn}_{0.125}\text{Fe}_{1.875}\text{O}_4$ , sample cobalt ferrite nanoparticles.**

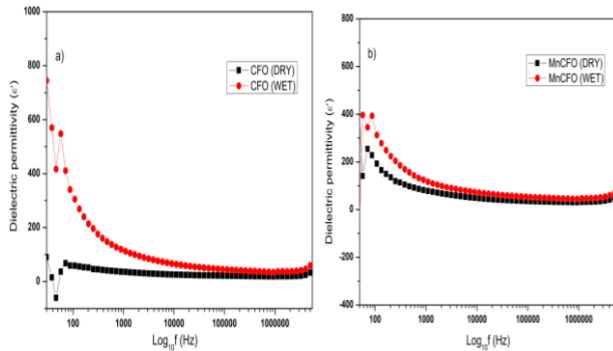
cobalt ferrite nanoparticles are affected by the particle size, production process, and cation dispersion. These elements produce surface disorder brought on by a canted spin structure and random cation distribution. The high surface disorder generally occurs due to the presence of high surface anisotropy [24]. CFO based compounds have been heavily researched because of its strong coercivity, great chemical stability, superior electrical insulation, significant mechanical hardness, and mild saturation magnetization at ambient temperature, etc [25].

**Table 3. Values of  $M_s$ ,  $M_r$ , and  $H_c$  for manganese doped cobalt ferrites (Mn-CFO)**

Chemical compound	$M_s$ (emu/g)	$M_r$ (emu/g)	$H_c$ (Oe)
CFO	48.258	26.502	1266.45
Mn-CFO	73.089	30.490	644.975

**4.6. Variation of dielectric constant ( $\epsilon'$ ) for dry and wet hydroelectric cells:**The ability of water dissociation by the Mn doped CFO based HEC and the conduction mechanism has been investigated using an impedance analyzer at room temperature. The variation of dielectric constant against frequency in the frequency range 20 Hz-  $10^6$

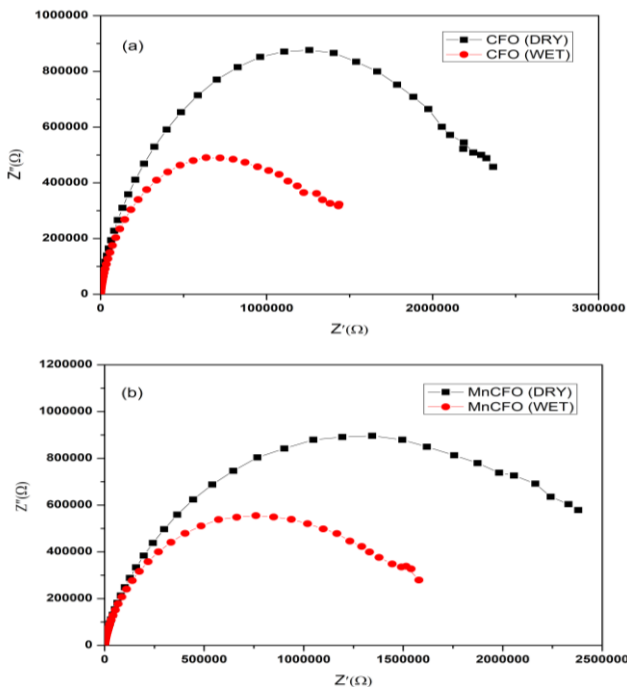
Hz is shown in Fig 8. It has been observed that the dielectric constant ( $\epsilon'$ ) values has been



**Fig 8. Dielectric constant with frequency for dry and wet hydroelectric cells of (a)  $\text{CoFe}_2\text{O}_4$  (b)  $\text{CoMn}_{0.125}\text{Fe}_{1.875}\text{O}_4$ .**

increased for wet HEC as compared to dry HEC at low frequency range. The  $\epsilon'$  values have become almost constant at high frequency range. This behaviour in the variation of  $\epsilon'$  values is mainly due to rise in ionic and space charge polarization due to dissociated water molecules [26].

**4.7. Ionic diffusion Plots:** EIS is a highly delicate technique to study the diffusion of ions on the material's surface. The various reactions occurring on the cell surface can be



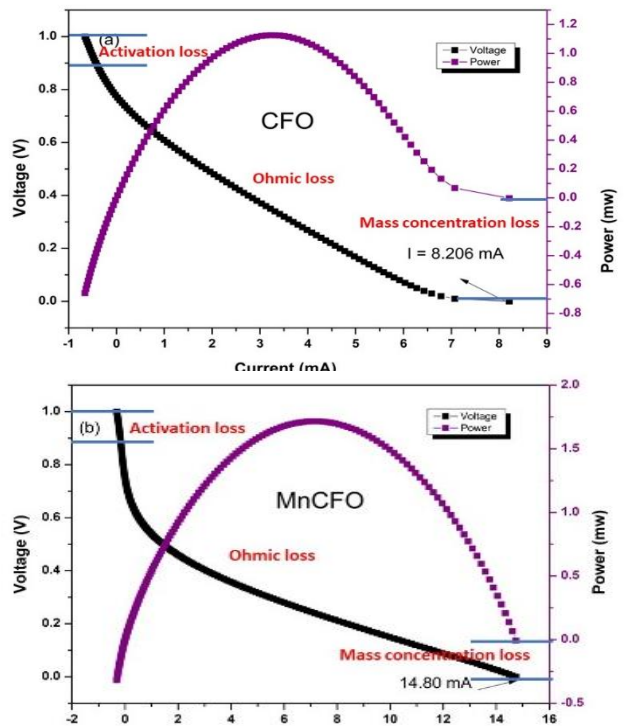
**Fig 9.  $Z''$  vs  $Z'$  plots for dry and wet hydroelectric cells of (a)  $\text{CoFe}_2\text{O}_4$  (b)  $\text{CoMn}_{0.125}\text{Fe}_{1.875}\text{O}_4$ .**

studied using EIS spectroscopy. The Nyquist's plots have shown the large impedance of dry

HEC in the order of 107- 108 ohms as compared to wet HEC. Wet HEC exhibit impedance in the range of few kilo ohms. This extreme deduction in the impedance value is due to the dissociation of water molecules into  $\text{H}_3\text{O}^+$  and  $\text{OH}^-$  ions. It confirms the water molecules had been adsorbed on the sample surface and got chemidissociated [27].

**4.8. Current and voltage studies in Mn doped CFO based hydroelectric cells:**

The variation of voltage against current for pure cobalt ferrite as well as Mn doped CFO is



**Fig 10. V-I polarization plots for dry and wet HEC of (a)  $\text{CoFe}_2\text{O}_4$  (b)  $\text{CoMn}_{0.125}\text{Fe}_{1.875}\text{O}_4$ .**

shown in Fig 10. The maximum peak off-load current ( $I_{sc}$ ) around 14.80 mA has been reported for Mn doped CFO based HEC. The high value of current ( $\sim 14.80$  mA) has been accommodated due to the production of more oxygen vacancies in Mn doped CFO. It has been observed that the concentration loss gets reduced for doped cells due to which current remains constant for a longer time and gives high value. The whole V-I polarization curve is divided into mainly three regions:- Activation loss, ohmic loss and mass concentration loss. Activation loss refers to the energy barrier for the charge transfer reaction

at both electrodes. Ohmic region is the region where voltage decreases in almost linear manner with the current. The sharp degradation in the voltage has been occurred due to the aggregation of ions near the electrodes and caused a concentration loss or mass loss [28,29].

**5. Conclusions:** A comprehensive investigation was conducted to check the impact of Mn substitution on the generation of current by hydroelectric cells of cobalt ferrites. The presence of  $Mn^{2+}$  ions in the lattice has led to the generation of strain and oxygen vacancies. X-ray diffraction pattern and Fourier transform spectroscopy has confirmed the formation of cubic phase of  $Co_{1-x}Mn_xFe_2O_4$  (where  $x = 0.00, 0.125$ ) ferrites. XRD studies revealed the net increment in the porosity % for the Mn doped CFO. Increased porosity initiated increased chemi-dissociation of water molecules. High values of saturation magnetization obtained at room temperature suggests the ferromagnetic nature of the material. The fitted Nyquist plot of Mn doped CFO based HEC shows a drop in both charge transfer resistance and ohmic resistance. This drop, in turn, enhanced the reaction rate and a corresponding enhancement has been observed in ionic conduction. This enhanced ionic conduction produces an output current of 14.80 mA, accompanied by voltage of 0.98 V, for the substitution of Mn in CFO. This study demonstrates the improved performance of hydroelectric cells based on transition metal doped ferrites.

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